## Fe-Mn-Tb (Iron-Manganese-Terbium)

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The previous review of this system by [1992Rag] was limited to a summary of the lattice parameter variation of the continuous solid solutions along the  $TbFe_2$ - $TbMn_2$  and  $Tb_6Fe_{23}$ - $Tb_6Mn_{23}$  joins. Ilyushin et al. [1994IIy] determined the phase relationships along the  $TbFe_2$ - $TbMn_2$  join as a function of pressure up to 8 GPa.

## **Binary Systems**

The Fe-Mn phase diagram [1993Oka] has no intermediate phases. The face-centered-cubic phases  $\gamma$ Fe and  $\gamma$ Mn form a continuous solid solution. In the Fe-Tb system [Massalski2], there are four intermediate phases: the Th<sub>2</sub>Ni<sub>17</sub>type hexagonal and the Th<sub>2</sub>Zn<sub>17</sub>-type rhombohedral modifications of Tb<sub>2</sub>Fe<sub>17</sub>, the Th<sub>6</sub>Mn<sub>23</sub>-type cubic phase Tb<sub>6</sub>Fe<sub>23</sub>, the PuNi<sub>3</sub>-type rhombohedral phase TbFe<sub>3</sub>, and the MgCu<sub>2</sub>-type cubic phase TbFe<sub>2</sub>. The Mn-Tb phase diagram [Massalski2] shows three intermediate phases: the ThMn<sub>12</sub>type tetragonal phase TbMn<sub>12</sub>, the Th<sub>6</sub>Mn<sub>23</sub>-type cubic phase Tb<sub>6</sub>Mn<sub>23</sub>, and the MgCu<sub>2</sub>-type cubic phase TbMn<sub>2</sub>.

## **Ternary Phase Equilibria**

With starting metals of purity of 99.99%, [1994Ily] levitation melted under Ar atm 13 alloy compositions along the TbFe<sub>2</sub>-TbMn<sub>2</sub> join. The samples were then powdered and subjected to a high pressure of 1-8.0 GPa. The structure of the synthesized alloys was determined at ambient temperature by x-ray powder diffraction. At atmospheric pressure, the structure of all compositions  $\text{Tb}(\text{Fe}_{1-x}\text{Mn}_x)_2$  is of the *C*15 cubic type. However, a detailed analysis of the x-ray patterns showed a rhombohedral distortion in the range  $x \sim$ 0 to 0.3. For the range  $x \sim 0.3$  to 0.5, the distortion becomes more complex (closer to monoclinic). For  $x \sim 0.5$  to 1.0, the alloys have the ideal cubic structure. At the TbMn<sub>2</sub> end, the cubic *C*15 phase is stable up to ~3 GPa of pressure. Above 6 GPa, the hexagonal *C*14 phase is stable. In between, a (*C*14 + *C*15) two-phase mixture prevails. The pressurecomposition diagram constructed by [1994IIy] at room temperature is shown in Fig. 1.

## References

- **1992Rag:** V. Raghavan: "Fe-Mn-Tb (Iron-Manganese-Terbium)" in *Phase Diagrams of Ternary Iron Alloys. Part 6*, Ind. Inst. Metals, Calcutta, India, 1992, p. 967.
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- 1994Ily: A.S. Ilyushin, I.A. Nikanorova, A.V. Tsvyashchenko, M.A.A. Gudaev, S. Lei, and Z. Guien: "Phase Diagram of the Quasi-Binary System Tb(Fe<sub>1-x</sub>Mn<sub>x</sub>)<sub>2</sub> Synthesized at High Pressures," *Vestn. Mosk. Univ. Ser. 3: Fiz. Astron.*, 1994, 35(1), pp. 101-02 (in Russian); TR: *Moscow Univ. Phys. Bull.*, 1994, 49(1), pp. 100-101.



Fig. 1 Fe-Mn-Tb pressure-composition diagram for  $Tb(Fe_{1-x}Mn_x)_2$  alloys at ~20 °C [1994Ily]